# Synthesis of the Quaternary One-Dimensional Triple-Chain Materials $K_3Cu_3M_2Q_8$ (M = Nb, Ta; Q = S, Se): The Structure of $K_3Cu_3Nb_2S_8$

# YING-JIE LU AND JAMES A. IBERS

Department of Chemistry, Northwestern University, Evanston, Illinois 60208-3113

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The one-dimensional triple-chain materials  $K_3Cu_3Nb_2S_8$ ,  $K_3Cu_3Nb_2S_8$ ,  $K_3Cu_3Ta_2S_8$ , and  $K_3Cu_3Ta_2S_8$ have been prepared by the reactive flux method in which Cu and Nb or Ta were reacted with  $K_2Q_5/Q$  (Q = S, Se) at 850°C.  $K_3Cu_3Nb_2S_8$  has been characterized by single-crystal X-ray diffraction techniques. It crystallizes with four formula units in space group  $C_{2h}^6-C2/c$  of the monoclinic system in a cell of dimensions a = 23.446(3), b = 5.545(3), c = 14.204(3) Å,  $\beta = 120.87(3)^\circ$  at  $-120^\circ$ C. It is composed of one-dimensional  $\frac{1}{6}$ [Cu<sub>3</sub>Nb<sub>2</sub>S<sup>3</sup><sub>8</sub>] chains isolated from one another by K<sup>-</sup> cations. These chains are triple in nature, and consist of successive corner-sharing CuS<sub>4</sub>-CuS<sub>4</sub> tetrahedra and edge-sharing CuS<sub>4</sub>-NbS<sub>4</sub> tetrahedra. This edge-sharing results in Nb-Cu distances of 2.772(2)-2.781(1) Å. K<sub>3</sub>Cu<sub>3</sub> Nb<sub>2</sub>Se<sub>8</sub> with cell dimensions a = 23.453(4), b = 5.592(12), c = 14.223(3) Å,  $\beta = 120.7(3)^\circ$  at  $-120^\circ$ C ( $\lambda$ (CuK $\alpha_1$ ) = 1.54056 Å); K<sub>3</sub>Cu<sub>3</sub>Ta<sub>2</sub>Se<sub>8</sub> with cell dimensions a = 23.493(9), b = 5.538(7), c = 14.204(12)Å,  $\beta = 120.5(9)^\circ$  at  $-165^\circ$ C ( $\lambda$ (MoK $\alpha_1$ ) = 0.70930 Å); and K<sub>3</sub>Cu<sub>3</sub>Ta<sub>2</sub>Se<sub>8</sub> with cell dimensions a = 23.450(6), b = 5.541(3), c = 14.110(6) Å,  $\beta = 120.47(3)^\circ$  at  $-165^\circ$ C ( $\lambda$ (MoK $\alpha_1$ ) = 0.70930 Å) appear to be isostructural with K<sub>3</sub>Cu<sub>3</sub>Nb<sub>2</sub>S<sub>8</sub>. K<sub>3</sub>Cu<sub>3</sub>Nb<sub>2</sub>S<sub>8</sub> has a resistivity greater than 10<sup>5</sup> m\Omega cm at room temperature. @ 1992 Academic Press, Inc.

# Introduction

Many new quaternary/K/Cu/Nb or Ta/ S or Se/solid-state compounds, most with unusual structures, have been synthesized by use of the reactive flux method (1). These include the one-dimensional chain compounds  $K_2CuMSe_4$  (M = Nb(2), Ta(3)) and  $K_3CuNb_2Se_{12}$  (2). These structures exhibit one-dimensional mixed-metal chains formed by successive edge-sharing of tetrahedra ( $K_2CuMSe_4$ ) or by alternation of facesharing and edge-sharing of mixed polyhedra and tetrahedra ( $K_3CuNb_2Se_{12}$ ). With the addition of CuSe<sub>4</sub> tetrahedra in a cornersharing arrangement, the two-dimensional layer compounds  $\text{KCu}_2M\text{Se}_4$  (M = Nb(4)),

Ta (3)) and KCu<sub>2</sub>NbS<sub>4</sub> (4) are formed. The conceptual addition of corner-sharing CuQ<sub>4</sub> (Q = S, Se) tetrahedra to the  $\frac{1}{\infty}$ [CuMQ<sub>4</sub><sup>2-</sup>] structure would produce a one-dimensional triple-chain structure. Here we report the synthesis and characterization of the K<sub>3</sub>Cu<sub>3</sub>  $M_2Q_8$  series, a series that exhibits this structure.

### Experimental

Synthesis.  $K_3Cu_3Nb_2S_8$  was prepared from a stoichiometric reaction of  $K_2S_5$ (142.8 mg, 0.6 mmol) with powders of elemental Nb (74 mg, 0.8 mmol), Cu (76 mg, 1.2 mmol), and S (6.4 mg, 0.2 mmol) (Nb, 99.8%, AESAR; Cu, 99.5%, Alfa; K, 99%, AESAR; S, 100.0%, Mallinckrodt). K<sub>2</sub>S<sub>5</sub> was prepared by the stoichiometric reaction of elemental K and S in liquid ammonia under an atmosphere of Ar. In a glove box the starting materials were loaded into a quartz tube that was subsequently evacuated to  $10^{-4}$  Torr and sealed. It was then placed in a furnace that was heated to 500°C, held there for 24 hr, then ramped to 850°C, kept there for four days, and then cooled to 25°C at a rate of 4°C/hr. Yellow, plate-like single crystals extracted from the surface of the melt were suitable for X-ray diffraction analysis. Chemical analysis of three crystals selected at random was performed with the electron microprobe of an EDAX-equipped Hitachi S-570 LB scanning electron microscope; the composition found K: Cu: Nb: Swas -----1.5:1.5:1:4.2,in good agreement with that of 3:3:2:8 established from the X-ray structure determination.

 $K_3Cu_3Nb_2Se_8$  was synthesized from a similar reaction of K<sub>2</sub>Se<sub>5</sub> (284 mg, 0.6 mmol) with elemental Nb (74 mg, 0.8 mmol), Cu (76 mg, 1.2 mmol), and Se (16 mg, 0.2 mmol). K<sub>2</sub>Se<sub>5</sub> was prepared by the stoichiometric reaction of K with Se in liquid ammonia. The heating regime was the same as above, but the cooling rate was 2°C/hr. Deep yellow and yellow-orange crystals extracted from the surface of the melt were suitable for X-ray diffraction analysis. Chemical analysis by EDAX gave K: Cu: Nb: Se = 1.8: 1.6: 1: 4.6.

 $K_3Cu_3Ta_2S_8$  was synthesized from a reaction of  $K_2S_5$  (100 mg, 0.42 mmol) with elemental Ta (101 mg, 0.56 mmol), Cu (53 mg, 0.84 mmol), and S (4.5 mg, 0.14 mmol) in the same heating regime used for  $K_3Cu_3Nb_2Se_8$ . Bright yellow and orange plate-like crystals that formed in the tube were suitable for X-ray analysis. Chemical analysis by EDAX gave K : Cu : Ta : S = 1.4 : 1.5 : 1 : 4.1.

 $K_3Cu_3Ta_2Se_8$  was synthesized from a reaction of  $K_2Se_5$  (284 mg, 0.6 mmol) with elemental Ta (145 mg, 0.8 mmol), Cu (76 mg, 1.2 mmol), and Se (16 mg, 0.2 mmol). In the same heating regime used for  $K_3Cu_3Nb_2Se_8$ , yellow-orange plate-like crystals formed in the tube that were suitable for X-ray analysis. Chemical analysis by EDAX gave K:Cu:Ta:Se = 1.6:1.7:1:4.4.

X-ray crystallography. Weissenberg photography indicated that K<sub>3</sub>Cu<sub>3</sub>Nb<sub>2</sub>S<sub>8</sub> belongs to the monoclinic system and displays systematic absences consistent with the space groups  $C_s^4 - Cc$ , and  $C_{2h}^6 - C2/c$ . The unit cell dimensions, a = 23.446(3), b = 5.545(3), c = 14.204(3) Å,  $\beta = 120.87(3)^{\circ}$  at  $-120^{\circ}$ C, were obtained by a least-squares analysis of the setting angles of 25 reflections automatically centered in the range of  $43^\circ \leq 2\theta$  $(CuK\alpha_1) \le 46^\circ$  on an Enraf-Nonius CAD4 diffractometer. K<sub>3</sub>Cu<sub>3</sub>Nb<sub>2</sub>Se<sub>8</sub> with cell dimensions a = 23.453(4), b = 5.592(12), c =14.223(3) Å,  $\beta = 120.7(3)^{\circ}$  at  $-120^{\circ}$ C  $(\lambda(CuK\alpha_1) = 1.54056 \text{ \AA}); \text{ K}_3\text{Cu}_3\text{Ta}_2\text{Se}_8 \text{ with}$ cell dimensions a = 23.493(9), b = 5.538(7),c = 14.204(12) Å,  $\beta = 120.5(9)^{\circ}$  at  $-165^{\circ}$ C  $(\lambda(MoK\alpha_1) = 0.70930 \text{ Å}); \text{ and } K_3Cu_3Ta_2S_8$ with cell dimensions a = 23.450(6), b =5.541(3), c = 14.110(6) Å,  $\beta = 120.47(3)^{\circ}$  at  $-165^{\circ}C (\lambda(MoK\alpha_{1}) = 0.70930 \text{ Å})$  appear to be isostructural. Intensity data for K<sub>3</sub>Cu<sub>3</sub>Nb<sub>2</sub>S<sub>8</sub> were collected by methods standard in this laboratory (5) with use of the  $\theta$ -2 $\theta$  technique in the range 2°  $\leq$  $\theta(CuK\alpha_1) \leq 75^\circ$ . Six standard reflections, chosen from diverse regions of reciprocal space and monitored every 3 hr during data collection, showed no significant change. In the solution and refinement of the structure all calculations were carried out on a Stellar GS2000 computer. The centrosymmetric space group  $C_{2h}^6 - C^2/c$  was chosen, since averaging *hkl* and  $h\bar{k}l$  reflections, after correction for absorption, led to a satisfactory residual of 0.069. The structure was solved by direct methods (6). An analytical absorption correction was applied (7). The structure was refined on  $F_0^2$  by full-matrix leastsquares methods. The final anisotropic re-

TABLE I Crystal Data and Experimental Details for K1Cu1Nb.Sy

Formula	K <sub>3</sub> Cu <sub>3</sub> Nb <sub>2</sub> S <sub>8</sub>
Formula weight	750.27
Space group	$C_{2h}^{6} - C_{2h}^{6} - C_{2h}^{6}$
a, Å	23.446(3)
b, Å	5.545(3)
<i>c</i> , Å	14.204(3)
$\beta$ , deg.	120.87(3)
Vol. Å <sup>3</sup>	1583
Ζ	4
Temperature, °C	$-120^{\alpha}$
$\rho(calc), g cm^{-3} (-120^{\circ}C)$	3.147
$\lambda(CuK\alpha_1), Å$	1.54056
$\mu,  \mathrm{cm}^{-1}$	330
Crystal shape	Plate bounded by $\{20\overline{1}\}$ ,
	{010}, {101}
Transmission factors	0.070-0.279
Crystal vol., mm <sup>3</sup>	$1.3 \times 10^{-3}$
Take-off angle, deg.	3.5
Scan type	$\theta - 2\theta$
Scan range, $\theta$ deg.	±1.0
Data collected	$2 \leq \theta \leq 75^\circ; \pm h, \pm k, l$
p factor	0.04
No. of data collected	3372
No. of unique data	1727
No. of unique data with	1403
$F_{\rm o}^2 > 3\sigma(F_{\rm o}^2)$	
$R \text{ [on } F \text{ for } F_0^2 > 3\sigma(F_0^2) \text{]}$	0.069
$R(F_0^2)$	0.152
$R_w(F_o^2)$	0.159
Error in observation of unit weight $(e^2)$	2.54
- • •	

<sup>a</sup> The low temperature system for the Nonius CAD4 diffractometer is from a design by Professor J. J. Bonnet and S. Askenazy and is commercially available from Soterem, Z. T. de Vic, 31320 Castanet-Tolosan, France.

finement of 1728 observations and 76 variables converged to values of  $R(F_0^2)$  of 0.152 and  $R_w(F_0^2)$  of 0.159. The conventional R index R(F) for those 1403 reflections having  $F_0^2 > 3\sigma(F_0^2)$  is 0.069. Some crystallographic details are given in Table I. The final positional and equivalent isotropic thermal parameters are given in Table II. Anisotropic thermal parameters and structure amplitudes are given in Table IS.<sup>1</sup>

<sup>1</sup> See NAPS document No. 04922 for 10 pages of supplementary materials. Order from ASIS/NAPS, Microfiche Publications, P.O. Box 3513, Grand Central Station, New York, NY 10163. Remit in advance \$4.00 for microfiche copy or for photocopy, \$7.75. All orders must be prepaid. Institutions and Organizations may order by purchase order. However, there is a billing



FIG. 1. Projection of the  $K_3Cu_3Nb_2S_8$  structure down [010].

## **Results and Discussion**

The crystal structure of  $K_3Cu_3Nb_2S_8$  consists of well-separated cations and anions (Fig. 1). The anion is the new one-dimensional chain  $\frac{1}{2}$ [Cu<sub>3</sub>Nb<sub>2</sub>S<sub>8</sub><sup>3–</sup>]. Figure 2 shows this anionic chain with the labeling scheme. This triple-metal chain, which runs along [010], comprises successive CuS<sub>4</sub>–NbS<sub>4</sub> edge-sharing tetrahedra and CuS<sub>4</sub>–CuS<sub>4</sub> corner-sharing tetrahedra. It may be thought of as the addition of CuQ<sub>4</sub>–CuQ<sub>4</sub> corner-sharing tetrahedra to the successive CuQ<sub>4</sub>–NbQ<sub>4</sub> chains in K<sub>2</sub>CuNbSe<sub>4</sub> (2). In the present structure there are two crystallographically distinct Cu atoms. The NbS<sub>4</sub> and Cu(2)S<sub>4</sub> tetrahedra edge share along *b*, while



FIG. 2. The one-dimensional triple chain  $\frac{1}{\kappa}[Cu_3Nb_2S_8^{3-}]$  in  $K_3Cu_3Nb_2S_8.$ 

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Atom	Wyckoff Notation	r	v	7	$B_{aa}^{a}$
			·	~	₽ eq
Nb(1)	8 <i>f</i>	0.127063(29)	0.28240(13)	0.280696(47)	0.73(2)
Cu(1)	4e	0	0.27823(37)	1	1.13(5)
Cu(2)	8 <i>f</i>	0.376968(65)	0.28201(26)	0.21811(10)	1.14(4)
K(I)	8f	0.30991(10)	0.03699(42)	0.40645(16)	1.76(6)
K(2)	4 <i>a</i>	0	0	0	1.43(8)
S(1)	8 <i>f</i>	0.092236(96)	0.04895(40)	0.37627(15)	0.99(6)
S(2)	8f	0.038720(98)	0.51802(39)	0.15839(16)	1.01(6)
S(3)	8f	0.15757(10)	0.03092(40)	0.18658(16)	1.11(6)
S(4)	8f	0.287670(99)	0.02891(40)	0.10776(16)	1.11(6)

TABLE II

POSITIONAL AND EQUIVALENT ISOTROPIC THERMAL PARAMETERS (Å<sup>2</sup>) FOR K<sub>3</sub>Cu<sub>3</sub>Nb<sub>2</sub>S<sub>8</sub>

<sup>a</sup>  $B_{eq} = (8\pi^2/3) \sum_i \sum_j U_{ij} a_i^* a_j^* \mathbf{a}_j \cdot \mathbf{a}_j.$ 

the NbS<sub>4</sub> and Cu(1)S<sub>4</sub> tetrahedra edge share along *a* (Fig. 3). This sharing alternates Nb and Cu atoms, as was found in K<sub>2</sub>CuNbSe<sub>4</sub> (2). As a result of this edge-sharing, the Cu(1)S<sub>4</sub> and Cu(2)S<sub>4</sub> tetrahedra cornershare. The Cu(1), Cu(2), S(1), and S(2) atoms thus form eight-member rings, with Cu(1)–S–Cu(2) angles ranging from 109.73(8)° to 110.76(9)°. Figure 3b shows a polyhedral representation of the anionic chains.

Table III lists selected distances and angles in K<sub>3</sub>Cu<sub>3</sub>Nb<sub>2</sub>S<sub>8</sub>, while Table IV com-



FIG. 3. (a) View of the anionic chain  $\frac{1}{8}$ [Cu<sub>3</sub>Nb<sub>2</sub>S<sub>3</sub><sup>3-</sup>] down [001]. (b) Polyhedral representation of the structure of K<sub>3</sub>Cu<sub>3</sub>Nb<sub>2</sub>S<sub>8</sub> down [001].

pares metrical data among related structures. The Cu-S distances in the present structure range narrowly from 2.333(2) to 2.360(2) Å, comparing well with those in  $\alpha$ - and  $\beta$ -[CuS<sub>4</sub>]<sup>*n*-</sup><sub>*n*</sub> (Cu–S, 2.298(4)–2.432(4) Å (8)) and that in  $Cu_3VS_4$  (2.285(14) Å (9)). The Nb-Cu distances of 2.772(2), 2.777(2), and 2.781(1) Å are shorter than the corresponding distances in K<sub>2</sub>CuNbSe<sub>4</sub> (2.873(1) Å (2)) and in KCu<sub>2</sub>NbSe<sub>4</sub> (2.831(1)-2.833(4)) $\dot{A}$  (4)), because of the substitution of S for Se. The Nb-S distances range from 2.264(2) to 2.307(2) Å and are comparable to those in  $K_3NbS_4$  (2.241(8)–2.257(8) Å (10)). In transition-metal sulfides or selenides Nb atoms are most often found in trigonal prismatic coordination, as in NbS<sub>2</sub> (11), most binary chalcogenides (12), the incommensurate double-layer structures  $(MS)_n TS_2$ , n = 1.08-1.19 (M = Sn, Pb, Bi, rare earth elements; T = Nb, Ta) (13-15), and Nb<sub>2</sub>Pd<sub>3</sub>Se<sub>8</sub> (16), or in polyhedral coordination, as in Nb<sub>4</sub>Se<sup>6-</sup><sub>22</sub> (17) and  $K_3CuNb_2Se_{12}$  (2). While Nb is found in tetrahedral coordination in the  $MQ_4^{3-}$ anions (M = Nb, Ta; Q = S, Se) (10, 18) as well as in  $Ba_6(NbS_4)(NbS_3O)_3$  (19), tetrahedral coordination of Nb seems unusually prevalent in the K/Cu/Nb/Q systems (2-4, 20). The two crystallographi-

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Nb–Cu(1)	2.781(1)	K(1)–S(1)	3.577(3)
Nb-Cu(2)	2.777(2)	K(1) - S(3)'	3.296(3)
Nb-Cu(2)'	2.772(2)	K(1) - S(3)	3.326(3)
Cu(1)-Cu(2)	3.871(2)	K(1) - S(3)	3.352(3)
Cu(1) - Cu(2)'	3.841(2)	K(1) - S(4)	3.499(3)
Nb-S(1)	2.307(2)	K(1) - S(4)	3.569(3)
Nb-S(2)	2.307(2)	K(1) - S(4)	3.180(3)
Nb-S(3)	2.286(2)	K(2) - 2S(1)	3.426(2)
Nb-S(4)	2.264(2)	K(2)-2S(2)	3.306(2)
Cu(1)-2S(1)	2.351(2)	K(2)-2S(2)	3.470(2)
Cu(1)-2S(2)	2.345(2)	K(2)-2S(3)	3.264(2)
Cu(2) - S(1)	2.346(2)	S(1)-Nb-S(2)	107.98(7)
Cu(2) - S(2)	2.360(2)	S(3)-Nb-S(1)	108.26(8)
Cu(2)-S(3)	2.349(2)	S(3) - Nb - S(2)	109.55(7)
Cu(2)-S(4)	2.333(2)	S(4) - Nb - S(1)	112.27(7)
$S(1)\cdots S(1)$	3.956(4)	S(4) - Nb - S(2)	108.32(8)
$S(1)\cdots S(2)$	3.732(3)	S(4) - Nb - S(3)	110.40(8)
$S(1)\cdots S(2)$	3.858(3)	S(1)-Cu(1)-S(1)	114.54(13)
$S(1)\cdots S(2)$	3.979(3)	2S(1)-Cu(1)-S(2)	105.27(7)
$S(1)\cdots S(3)$	3.722(3)	2S(1)-Cu(1)-S(2)	110.47(7)
$S(1)\cdots S(3)$	3.878(3)	S(2)-Cu(1)-S(2)	110.91(13)
$S(1)\cdots S(4)$	3.796(3)	S(1)-Cu(2)-S(2)	115.47(9)
$S(1)\cdots S(4)$	3.955(3)	S(1)-Cu(2)-S(3)	104.89(9)
$S(2)\cdots S(2)$	3.862(4)	S(3)-Cu(2)-S(2)	109.94(9)
$S(2)\cdots S(2)$	3.899(4)	S(4)-Cu(2)-S(1)	115.40(9)
$S(2)\cdots S(3)$	3.752(3)	S(4)-Cu(2)-S(2)	104.30(9)
$S(2)\cdots S(3)$	3.856(3)	S(4)-Cu(2)-S(3)	106.54(9)
$S(2)\cdots S(4)$	3.705(3)	Cu(2)-Nb-Cu(1)	88.40(5)
$S(3)\cdots S(4)$	3.736(3)	Cu(2) - S(1) - Cu(1)	109.73(8)
$S(3) \cdots S(4)$	3.750(3)	Cu(2)-S(2)-Cu(1)	110.76(9)
$S(3) \cdots S(4)$	3.752(3)		
$S(4)\cdots S(4)$	3.599(4)		

TABLE III SELECTED DISTANCES (Å) AND ANGLES (deg) FOR K<sub>3</sub>Cu<sub>3</sub>Nb<sub>2</sub>S<sub>8</sub>

cally distinct Cu atoms show similar
distortions from tetrahedral symmetry with
S-Cu(1)-S angles ranging from 105.27(7)°
to 114.54(13)° and S-Cu(2)-S angles rang-
ing from 104.30(9)° to 115.47(9)°, while
the NbS <sub>4</sub> tetrahedra are somewhat more
regular ( $(S-Nb-S = 107.98(7)^{\circ}-112.27(7)^{\circ})$ .

There are four crystallographically distinct S atoms. Atoms S(3) and S(4) are connected to the Nb and Cu(2) atoms; atoms S(1) and S(2) are connected to all three metals. There are two crystallographically distinct K atoms. If we choose 4.0 Å as an upper limit for K-S interactions then atom K(1) is seven coordinate while atom K(2) is eight coordinate.

Since there are no close S...S interactions formal oxidation states of K(I), Cu(I), Nb(V), and S(-II) may be assigned. This assignment is consistent with the fact that the resistivity of K<sub>3</sub>Cu<sub>3</sub>Nb<sub>2</sub>S<sub>8</sub> is greater than  $10^5 \text{ m}\Omega$  cm at room temperature.

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С

			·	
Compound	Cu–Q	Nb-Q	Nb-Cu	Ref.
$\alpha$ - and $\beta$ -KCuS <sub>4</sub>	2.298(4)-2.432(4)			(8)
Cu <sub>3</sub> VS <sub>4</sub>	2.285(14)			(9)
K <sub>3</sub> NbS <sub>4</sub>		2.241(8)-2.257(8)		(10)
$Ba_6(NbS_4)(NbS_3O)_3$		2.238(5) - 2.284(5)		(19)
KCu <sub>3</sub> S <sub>2</sub>	2.24-2.86			(21)
KCu <sub>4</sub> S <sub>3</sub>	2.312(2) - 2.451(1)			(22)
[NH <sub>4</sub> ][CuMoS <sub>4</sub> ]	2.283(4)			(23)
CuAsS	2.30(1) - 2.34(1)			(24)
TlCu <sub>7</sub> Se₄	2.377(5)-2.658(3)			(25)
TlCu <sub>5</sub> Se <sub>3</sub>	2.389(2)-2.629(3)			(26)
K <sub>2</sub> CuNbSe₄	2.457(1)	2.413(1)	2.873(1)	(2)
KCu <sub>2</sub> NbSe <sub>4</sub>	2.427(2) - 2.440(4)	2.392(3)-2.452(3)	2.831(1)-2.833(4)	(4)
$K_3Cu_3Nb_2S_8$	2.333(2)-2.360(2)	2.264(2)-2.307(2)	2.772(2)-2.781(1)	This work

TABLE IV

COMPARATIVE DISTANCES (Å) IN RELATED  $MO_4$  TETRAHEDRA (M = Cu, Nb; Q = S, Se)

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